inorganic compounds

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Rietveld refinement of KLaTiO₄ from X-ray powder data

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Key indicators: powder X-ray study; T = 298 K; mean σ (La–O) = 0.003 Å; R factor = 0.046; wR factor = 0.068; data-to-parameter ratio = 98.0.

Potassium lanthanum titanate(IV), KLaTiO₄, has been synthesized by conventional solid-state reaction. It crystallizes isotypically with the NaLnTiO₄ (Ln = La, Pr, Nd, Sm, Eu, Gd, Y and Lu) family. Five of the six atoms in the asymmetric unit (one K, one La, one Ti and two O atoms) are situated on sites with 4mm symmetry, whereas one O atom has 2mm. site symmetry. The crystal structure can be described as being composed of single layers of distorted corner-sharing TiO₆ octahedra extending parallel to (001). The layers are alternately separated by K^+ and La^{3+} cations along [001]. The coordination number of both K⁺ and La³⁺ cations is nine, resulting in distorted KO₉ and LaO₉ polyhedra.

Related literature

For the isotypic NaLnTiO₄ (Ln = La, Pr, Nd, Sm, Eu, Gd, Yand Lu) family, see: Toda et al. (1996a). Orthorhombic symmetry for other members of this family has been reported by Nishimoto et al. (2006). Decomposition products of $NaLnTiO_4$ were investigated by Toda *et al.* (1996b). For preparation by ion-exchange and structure analysis of $KLnTiO_4$ (Ln = La, Nd, Sm, Eu, Gd, Dy) compounds, see: Schaak & Mallouk (2001). For hydrothermal preparation of similar compounds, see: Dairong et al. (1999). For crystallographic background, see: Howard (1982); Thompson et al. (1987).

(10) Å

Experimental

Crystal data

a = 3.84155 (10) Å
c = 13.4695 (4) Å
V = 198.78 (1) Å ³

Z = 2Cu $K\alpha$ radiation, $\lambda = 1.54060$, 1.54443 Å

Data collection

PANalytical X'pert PRO diffractometer Specimen mounting: packed powder pellet	Data collection mode: reflection Scan method: continuous $2\theta_{\min} = 9.872^{\circ}, 2\theta_{\max} = 109.815^{\circ}, 2\theta_{step} = 0.017^{\circ}$
Refinement	
$R_{\rm p} = 0.046$	$\chi^2 = 2.220$
$R_{wp} = 0.068$	5880 data points
$R_{\rm exp} = 0.046$	60 parameters
$R(\dot{F}^2) = 0.047$	-

T = 298 K

flat sheet, 20 \times 20 mm

Table 1 Selected bond lengths (Å).

K1-O1	3.065 (4)	La1-O3 ⁱ	2.7628 (12)
K1-O2	2.765 (9)	Ti1-O1	1.9635 (12)
$K1 - O2^i$	2.7242 (7)	Ti1-O2 ⁱⁱⁱ	1.775 (9)
La1-O1	2.530 (3)	Ti1-O3 ⁱⁱⁱ	2.558 (7)
La1-O3 ⁱⁱ	2.339 (7)		

Symmetry codes: (i) -x, -y, -z + 1; (ii) x, y, z + 1; (iii) -x + 1, -y + 1, -z + 1.

Data collection: X'pert Data Collector (PANalytical, 2003); cell refinement: GSAS (Larson & Von Dreele, 2004) and EXPGUI (Toby, 2001); data reduction: X'pert Data Collector; method used to solve structure: coordinates taken from an isotypic compound (Toda et al., 1996b); program(s) used to refine structure: GSAS and EXPGUI; molecular graphics: DIAMOND (Brandenburg, 1999); software used to prepare material for publication: publCIF (Westrip, 2010).

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Supplementary data and figures for this paper are available from the IUCr electronic archives (Reference: WM2446).

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supplementary materials

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Rietveld refinement of KLaTiO₄ from X-ray powder data

B.-C. Zhu and K.-B. Tang

Comment

Layered perovskites that belong to the Ruddlesden-Popper $A'_2[A_{n-1}B_2O_{3n+1}]$ familiy (A' = alkali, A = alkaline earth or rare earth cation; <math>B = transition metal cation) possess a variety of interesting properties, such as superconductivity, colossal magnetoresistance, ferroelectricity, as well as catalytic activity. The structure of KLaTiO₄ we report here is a n = 1 member of this familiy. Isotypic crystal structures have been reported for NaLnTiO₄ (Ln = La, Pr, Nd, Sm, Eu, Gd, Y and Lu; Toda *et al.*, 1996*a*) in the space group P4/nmm.

Schaak & Mallouk (2001) reported the KLnTiO₄ (Ln= La, Nd, Sm, Eu, Gd, Dy) family of compounds to crystallize in space group *Pbcm*, as determined from Rietveld refinements of X-ray powder data. We tested both *Pbcm* and *P4/nmm* space groups with the underlying structures KLnTiO₄ (*P4/nmm*; Schaak & Mallouk, 2001) and NaLnTiO₄ (*P4/nmm*; Toda *et al.*, 1996*a*) as starting models for Rietveld refinement of KLaTiO₄. The results revealed the *P4/nmm* model to be significantly better than the *Pbcm* model. It is well-know that different rare earth elements can affect the crystal structure dramatically. In single layer Ruddlesden-Popper phase perovskites some studies reported that NaLnTiO₄ compounds have tetragonal symmetry for Ln = La—Nd, while an orthorhombic symmetry is observed for Ln = Sm—Lu and Y (Nishimoto *et al.*, 2006). We can infer that a similar situation might be present for KLnTiO₄ compounds. We ascribe the difference in symmetry between KLaTiO₄ obtained through solid state reactions (tetragonal) and through ion-exchange (orthorhombic) to the different temperature treatment (higher temperatures for the solid state reaction route).

Other methods used to prepare KLaTiO₄ have been reported previously, like an ion exchange method by Schaak & Mallouk (2001) and a hydrothermal method by Dairong *et al.* (1999). To our knowledge, a solid state route to synthesize this compound and its detailed structure analysis based on Rietveld refinement from X-ray powder diffraction data has not been reported. KLaTiO₄ easily decomposes at high temperature and is converted into the three-layer Ruddlesden-Popper phase $K_2La_2Ti_3O_{10}$. This phenomenon is also found in during preparation of NaLaTiO₄ reported by Toda *et al.* (1996*b*). Therefore we modified the reaction conditions on the basis of the preparation of NaLaTiO₄ and obtained a single phase product successfully.

Fig. 1 shows the observed difference plots (calculated, observed) of the Rietveld refinement.

Fig. 2 illustrates the structure of KLaTiO₄. It consists of a single layer of corner-sharing distorted TiO₆ octahedra extending parallel to (001). The layers are separated by alternating layers of K⁺ and La³⁺ cations along [001]. The TiO₆ octahedra (4mm symmetry) are considerably distorted. They have four equal equatorial Ti—O distances [1.9635 (12) Å], one very short Ti—O distance [1.775 (9) Å] toward the K layer and a significantly longer Ti—O distance [2.558 (7) Å] towards the La layer. The corresponding coordination polyhedra around the K⁺ and La³⁺ cations are distorted KO₉ and LaO₉ polyhedra, each with 4mm symmetry.

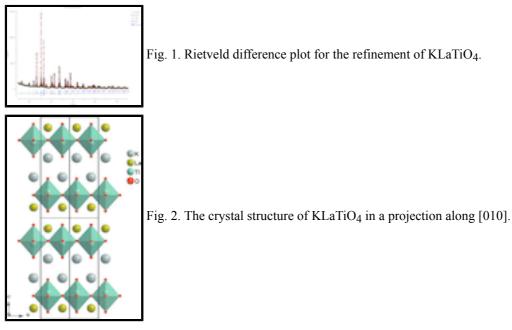
Experimental

The sample was prepared by conventional solid-state reaction. The starting materials were KNO_3 , La_2O_3 and TiO_2 in a molar ratio of 2:1:2. An excess of KNO_3 (55 mol%) was added to compensate for the loss due to the volatilization of the potassium component. La_2O_3 was heated to 1173 K for 10 h prior to use to remove water and carbonate impurities. The mixture was then ground and calcined at 1223 K for 30 min.

Refinement

The crystal structure of NaLaTiO₄ (Toda *et al.*, 1996*b*) in the spacegroup *P*4/*nmm* was used as a starting model for the final Rietveld refinement of the KLaTiO₄ structure. Isotropic displacement parameters were used for all atoms. The March-Dollase option in the *EXPGUI* program (Toby, 2001) was applied to correct for preferential orientation along [00*l*] which is often observed for such layered perovskites.

Figures



Potassium lanthanum titanate

Crystal data KLaTiO₄ $M_r = 289.90$ Tetragonal, P4/nmm Hall symbol: -p 4a 2a a = 3.84155 (10) Å c = 13.4695 (4) Å

Z = 2 $D_x = 4.848$ Mg m⁻³ Cu Kα radiation, $\lambda = 1.54060$, 1.54443 Å T = 298 K white flat sheet, 20 × 20 mm

$V = 198.78 (1) \text{ Å}^3$	Specimen preparation: Prepared at 1223 K
Data collection	
PANalytical X'pert PRO diffractometer Radiation source: sealed tube graphite Specimen mounting: packed powder pellet	Data collection mode: reflection Scan method: continuous $2\theta_{min} = 9.872^{\circ}, 2\theta_{max} = 109.815^{\circ}, 2\theta_{step} = 0.017^{\circ}$
Refinement	
Refinement on F^2	Excluded region(s): none Profile function: CW Profile function number 2 with 18 terms Profile coefficients for Simpson's rule integ- ration of pseudovoigt function C.J. Howard (1982). J. Appl. Cryst.,15,615-620. P. Thompson, D.E. Cox & J.B. Hastings (1987). J. Appl. Cryst.,20,79-83. #1(GU) = 0.000 #2(GV) = -2.261 #3(GW) = -9.290
Least-squares matrix: full	#4(LX) = 4.310 #5(LY) = 17.630 #6(trns) = 0.000 #7(asym) = 3.5282 #8(shft) = 0.0000 #9(GP) = 17.284 #10(stec) = 0.00 #11(ptec) = 0.00 #12(sfec) = 0.00 #13(L11) = 0.000 #14(L22) = 0.000 #15(L33) = 0.000 #16(L12) = 0.000 #17(L13) = 0.000 #18(L23) = 0.000 Peak tails are ignored where the intensity is below 0.0010 times the peak Aniso. broadening axis 0.0 0.0 1.0
$R_{\rm p} = 0.046$	60 parameters
$R_{\rm wp} = 0.068$	0 restraints
$R_{\rm exp} = 0.046$	$w = 1/[\sigma^{2}(F_{o}^{2}) + (0.0677P)^{2}]$ where $P = (F_{o}^{2} + 2F_{c}^{2})/3$
$R(F^2) = 0.04713$	$(\Delta/\sigma)_{\rm max} = 0.01$
$\chi^2 = 2.220$	Background function: GSAS Background function number 1 with 36 terms. Shifted Chebyshev func- tion of 1st kind 1: 353.285 2: -361.136 3: 220.846 4: -104.260 5: 61.8271 6: -33.1030 7: 19.7877 8: - 5.01446 9: 3.42337 10: -3.14370 11: 0.340114 12: 2.15882 13: -0.130836 14: -1.88421 15: 5.08631 16: -1.48077 17: 4.42719 18: 2.91556 19: -3.924060E- 0220: 0.679453 21: 5.77738 22: -2.47188 23: 3.81643 24: 3.21357 25: -4.71396 26: -1.63350 27: 0.665874 28: -7.16378 29: -7.040150E-0230: 3.04932 31: -2.36381 32: 0.787399 33: 4.27144 34: -2.96952 35: 4.90415 36: 1.54599
5880 data points	Preferred orientation correction: March-Dollase AX- IS 1 Ratio= 0.96438 h= 0.000 k= 0.000 l= 1.000 Prefered orientation correction range: Min= 0.94706, Max= 1.11492
Fractional atomic coordinates and isotropic o	r equivalent isotropic displacement parameters (A^2)

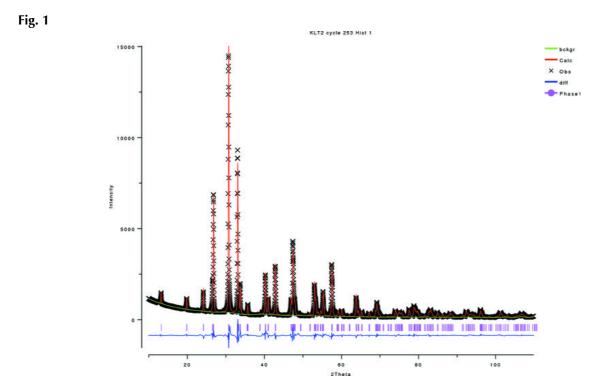
	x	У	Ζ	$U_{\rm iso}*/U_{\rm eq}$
K1	0.25	0.25	0.5950 (2)	0.0278 (9)*

supplementary materials

LA1	0.25	0.25	0.89446 (6)	0.0199 (4)*	
TI1	0.75	0.75	0.74203 (19)	0.0151 (7)*	
01	0.75	0.25	0.7723 (4)	0.0189 (17)*	
O2	0.25	0.25	0.3897 (6)	0.039 (2)*	
O3	0.25	0.25	0.0681 (5)	0.018 (2)*	
Geometric parame	ters (Å, °)				
K1—O1 ⁱ		3.065 (4)	La1—O1 ⁱⁱⁱ		2.5295 (32)
K1—O1		3.065 (4)	La1—O3 ^{viii}		2.339 (7)
K1—O1 ⁱⁱ		3.065 (4)	La1—O3 ^{iv}		2.7628 (12)
K1—O1 ⁱⁱⁱ		3.065 (4)	La1—O3 ^v		2.7628 (12)
K1—O2		2.765 (9)	La1—O3 ^{vi}		2.7628 (12)
K1—O2 ^{iv}		2.7242 (7)	La1—O3 ^{vii}		2.7628 (12)
$K1 - O2^{v}$		2.7242 (7)	Til—O1		1.9635 (12)
K1—O2 ^{vi}		2.7242 (7)	Ti1—O1 ^{ix}		1.9635 (12)
K1—O2 ^{vii}		2.7242 (7)	Ti1—O1 ⁱⁱⁱ		1.9635 (12)
La1—O1 ⁱ		2.5295 (32)	Ti1—O1 ^x		1.9635 (12)
La1—O1		2.5295 (32)	Ti1—O2 ^{vii}		1.775 (9)
La1—O1 ⁱⁱ		2.5295 (32)	Ti1—O3 ^{vii}		2.558 (7)
01 ⁱ —K1—O1		77.62 (13)	O1 ⁱ —La1—O3 ^v		65.85 (12)
01 ⁱ —K1—O1 ⁱⁱ		52.61 (8)	O1 ⁱ —La1—O3 ^{vi}		130.30 (11)
01 ⁱ —K1—O1 ⁱⁱⁱ		52.61 (8)	O1 ⁱ —La1—O3 ^{vii}		130.30 (11)
O1 ⁱ —K1—O2		141.19 (6)	O1—La1—O1 ⁱⁱ		64.95 (9)
01 ⁱ —K1—O2 ^{iv}		59.94 (15)	O1—La1—O1 ⁱⁱⁱ		64.95 (9)
$O1^{i}$ —K1— $O2^{v}$		59.94 (15)	O1—La1—O3 ^{viii}		130.59 (9)
01 ⁱ —K1—O2 ^{vi}		112.52 (18)	O1—La1—O3 ^{iv}		130.30 (11)
01 ⁱ —K1—O2 ^{vii}		112.52 (18)	O1—La1—O3 ^v		130.30 (11)
01—K1—01 ⁱⁱ		52.61 (8)	O1—La1—O3 ^{vi}		65.85 (12)
01—K1—01 ⁱⁱⁱ		52.61 (8)	O1—La1—O3 ^{vii}		65.85 (12)
O1—K1—O2		141.19 (6)	O1 ⁱⁱ —La1—O1 ⁱⁱⁱ		98.81 (17)
O1—K1—O2 ^{iv}		112.52 (18)	O1 ⁱⁱ —La1—O3 ^{viii}		130.59 (9)
O1—K1—O2 ^v		112.52 (18)	O1 ⁱⁱ —La1—O3 ^{iv}		65.85 (12)
O1—K1—O2 ^{vi}		59.94 (15)	O1 ⁱⁱ —La1—O3 ^v		130.30 (11)
O1—K1—O2 ^{vii}		59.94 (15)	O1 ⁱⁱ —La1—O3 ^{vi}		65.85 (12)
O1 ⁱⁱ —K1—O1 ⁱⁱⁱ		77.62 (13)	O1 ⁱⁱ —La1—O3 ^{vii}		130.30 (11)
O1 ⁱⁱ —K1—O2		141.19 (6)	O1 ⁱⁱⁱ —La1—O3 ^{viii}		130.59 (9)
$O1^{ii}$ —K1— $O2^{iv}$		59.94 (15)	O1 ⁱⁱⁱ —La1—O3 ^{iv}		130.30 (11)
$O1^{ii}$ —K1— $O2^{v}$		112.52 (18)	O1 ⁱⁱⁱ —La1—O3 ^v		65.85 (12)
O1 ⁱⁱ —K1—O2 ^{vi}		59.94 (15)	O1 ⁱⁱⁱ —La1—O3 ^{vi}		130.30 (11)
O1 ⁱⁱ —K1—O2 ^{vii}		112.52 (18)	O1 ⁱⁱⁱ —La1—O3 ^{vii}		65.85 (12)
O1 ⁱⁱⁱ —K1—O2		141.19 (6)	O3 ^{viii} —La1—O3 ^{iv}		79.48 (14)

$O1^{iii}$ —K1— $O2^{iv}$	112.52 (18)	O3 ^{viii} —La1—O3 ^v	79.48 (14)
$O1^{iii}$ —K1— $O2^{v}$	59.94 (15)	O3 ^{viii} —La1—O3 ^{vi}	79.48 (14)
O1 ⁱⁱⁱ —K1—O2 ^{vi}	112.52 (18)	O3 ^{viii} —La1—O3 ^{vii}	79.48 (14)
O1 ⁱⁱⁱ —K1—O2 ^{vii}	59.94 (15)	O3 ^{iv} —La1—O3 ^v	88.09 (5)
O2—K1—O2 ^{iv}	94.34 (18)	O3 ^{iv} —La1—O3 ^{vi}	88.09 (5)
$O2-K1-O2^{v}$	94.34 (18)	O3 ^{iv} —La1—O3 ^{vii}	158.96 (27)
O2—K1—O2 ^{vi}	94.34 (18)	O3 ^v —La1—O3 ^{vi}	158.96 (27)
O2—K1—O2 ^{vii}	94.34 (18)	O3 ^v —La1—O3 ^{vii}	88.09 (5)
$O2^{iv}$ —K1— $O2^{v}$	89.672 (27)	O3 ^{vi} —La1—O3 ^{vii}	88.09 (5)
$O2^{iv}$ —K1— $O2^{vi}$	89.672 (27)	O1—Ti1—O1 ^{ix}	156.05 (33)
$O2^{iv}$ —K1— $O2^{vii}$	171.3 (4)	O1—Ti1—O1 ⁱⁱⁱ	87.53 (7)
$O2^{v}$ —K1— $O2^{vi}$	171.3 (4)	$O1$ —Ti1— $O1^x$	87.53 (7)
$O2^{v}$ —K1— $O2^{vii}$	89.672 (27)	O1—Ti1—O2 ^{vii}	101.97 (16)
O2 ^{vi} —K1—O2 ^{vii}	89.672 (27)	O1 ^{ix} —Ti1—O1 ⁱⁱⁱ	87.53 (7)
Ol ⁱ —La1—O1	98.81 (17)	$O1^{ix}$ —Ti1— $O1^{x}$	87.53 (7)
O1 ⁱ —La1—O1 ⁱⁱ	64.95 (9)	O1 ^{ix} —Ti1—O2 ^{vii}	101.97 (16)
O1 ⁱ —La1—O1 ⁱⁱⁱ	64.95 (9)	O1 ⁱⁱⁱ —Ti1—O1 ^x	156.05 (33)
O1 ⁱ —La1—O3 ^{viii}	130.59 (9)	O1 ⁱⁱⁱ —Ti1—O2 ^{vii}	101.97 (16)
O1 ⁱ —La1—O3 ^{iv}	65.85 (12)	O1 ^x —Ti1—O2 ^{vii}	101.97 (16)
Symmetry codes: (i) $x-1 = y = 7$ (ii) $-y+$	1/2 r-1 z (iii) $-v+1/2$ r z	$(iv) - r - v - z + 1 \cdot (v) - r - v + 1 - z + 1 \cdot (v)$	v_{i} - r_{+1} - v_{-z+1} (v_{i}

Symmetry codes: (i) *x*-1, *y*, *z*; (ii) -*y*+1/2, *x*-1, *z*; (iii) -*y*+1/2, *x*, *z*; (iv) -*x*, -*y*, -*z*+1; (v) -*x*, -*y*+1, -*z*+1; (vi) -*x*+1, -*y*, -*z*+1; (vii) -*x*+1, -*y*, -*z*+1; (viii) *x*, *y*, *z*+1; (ix) *x*, *y*+1, *z*; (x) -*y*+3/2, *x*, *z*.



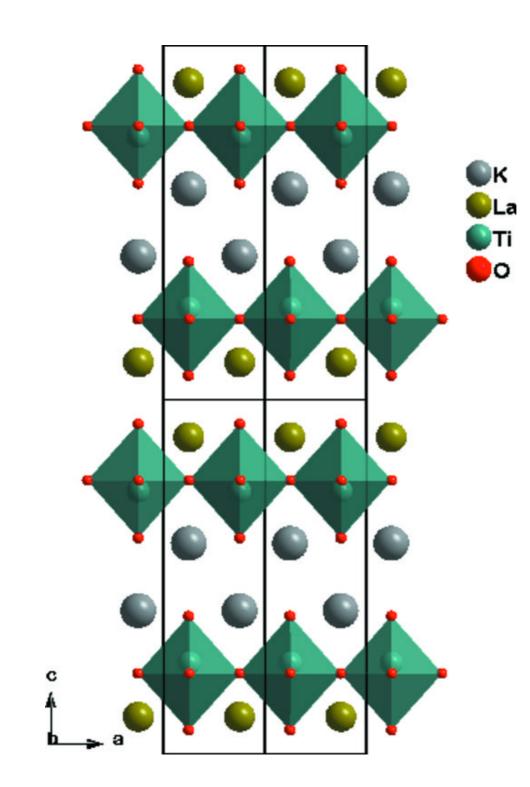


Fig. 2